## **Modeling of mechanical behavior**

In the numerical models (d), building upon models previously developed for wood [7,8], we study fiber-matrix interactions, modeling all possible failure mechanisms: matrix, interface, and fiber failure. We also explored the phase-field method [9] to improve the stability of nonlinear matrix behavior analysis. Representative volume elements for fiber-reinforced composites were developed to efficiently predict larger structures' mechanical behavior. The model was validated

We use two modeling strategies: continuum micromechanical models and detailed numerical models for nonlinear effects, like fiber-matrix interactions. For the former, we developed a model to predict plant fibers' stiffness and elastic limits [5]. Although the mechanical properties of these fibers vary widely, they all share basic components, with cellulose being the primary factor influencing mechanical behavior. Through continuum micromechanical multiscale modeling, the mechanical behavior of cellulose nanofibrils was upscaled to the technical fiber level (a). The predicted stiffness and elastic limits matched experimental values (b), validating the model and confirming that plant-specific physicochemical properties, such as microfibril angle and cellulose content, determine fiber performance. This model was further developed for plant-fiber-reinforced composites [6], including a spring-interface model for quantifying the compliance of the fibermatrix connection and the ability to model any orientation and length ratio distributions, and allowing for various parameter studies (c).

through comparison with existing tensile test data, with sensitivity analyses emphasizing the importance of matrix-fiber interface parameters. Due to the computational intensity, we aim to develop a multiscale approach that integrates the numerical model with the analytical micromechanics-based homogenization, e.g., by employing so-called deep material network models.

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### **Motivation and concept |**

 $\Omega$  During the production of lumber, a large proportion of by-products with different particle sizes is generated [1,2]. ② A combination of chemical and mechanical pretreatment is used to obtain (3) accessible and highly reactive fibers whose ligno-cellulosic composition can be tuned without destroying the inherent microstructure [3].  $\textcircled{4}$  The extracted components with high reactivity and cross-linking capabilities (also from underutilized parts of trees) are used as a binder in the  $(5)$  re-assembly stage [4]. There the formation of bonds leads to a  $\left(6\right)$  homogeneous and high-performant biocomposite material, which can be used in additive manufacturing processes. ا ا است ا $\sigma$  ال $\sigma$ 

> Mechanical treatment, leading to separation of fibers  $\blacksquare$

**How can the mechanical behavior be described as a function of material**  characteristics at these different length scales, using multiscale modeling and **simulation approaches?**

Nowadays, modeling concepts for biocomposites cannot describe all necessary mechanical processes on all relevant length scales. In contrast, often only basic models are used, which are supported by experimental campaigns.

In our modeling strategy, we combine fast analytical methods with very detailed nonlinear finite-element-based simulations to describe the mechanical behavior of our new wood-based biocomposite on all length scales. This approach is complemented by the development of new identification experiments on multiple length scales.

With this approach and together with detailed analysis techniques, it is possible to specifically optimize a biocomposite material by providing rapid feedback on the material behavior. In this way, the decisive process parameters in the material development can be identified and optimized, and the material models can be used to study biocomposite-based structural elements before the actual production phase.

# **Simulation-guided development concept for woodbased composites from sawmill byproducts**

**M. Lukacevic, M. Königsberger, G. Unsinn, M. Schwaighofer, V. Senk, L. Scolari, C. Hofbauer, N. Wahab, R. Ibadov, L. Zelaya-Lainez, S. Serna-Loaiza, T. Harter, M.Harasek, J. Füssl**

Hot-pressing of activated and impregnated ligno-

cellulosic material, leading to bond formation

 $\sqrt{1/\lambda}$  and impregnated and impregnated and impregnated and impregnated ligno-

 $\left[\begin{array}{cc} \sqrt{2} & \frac{1}{2} \\ \frac{1}{2} & \frac{1}{2} \end{array}\right]$  or  $\frac{1}{2}$  or

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*𝑓𝑖 𝝈<sup>𝑖</sup>* = *𝜮* (9)

∑

(a) The stiffness and viscoelastic properties of hot-pressed lignins were studied using nanoindentation of polished specimens [10]. A statistical and microstructure-guided evaluation using microscopy images allowed determining reliable properties of porous lignin. The desired mechanical properties of pure lignin were back-calculated using a continuum micromechanics model, indicating an intrinsic elastic modulus of approximately 7 GPa [11] and showing that lignin is a viscoelastic material with a pronounced short-term relaxation behavior during tests [12]. (b) Single-fiber tests are used to further validate the multiscale models. (c) Various lap-shear and tensile tests with veneers or specially produced papers (dynamic sheet former) have already been investigated, in which all production steps can be replicated and thus provide a good basis for comparison with and further development of the biocomposite material.

 $\delta_{\text{f}}^{\text{F}}$ f

(2)  $\delta^{\rm F}_{\rm c}$  (3)

 $\delta^\text{IF}_\text{I,f}$ i,f

 $\delta^\text{IF}_{i,(}$ i,0

*2.3. Intrinsic phase stiffnesses*

Chemical treatment of the construction and the chemical of the material of the hemicelluloses, plus possible activation of fiber structure Chemical treatment, leading to (partial) removal of lignin and hemicelluloses, plus possible activation of fiber structure



We are stiffness upscaling from the nanoscaling from the nanoscaling from the nanoscaling from the nanoscale, where  $\alpha$ 

*properties*





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*𝑁*

*𝑓𝑖*

*𝜺<sup>𝑖</sup>* − *𝜺𝑖*∕0

<sup>A</sup>*𝑖*(*𝜗, 𝜑*; *<sup>𝑎</sup>*) = <sup>A</sup>0*,𝑖*(*𝜗, 𝜑*; *<sup>𝑎</sup>*) ∶



*𝑓𝑖* ∫

*𝑓𝑎*(*𝑎*)

∫

<sup>0</sup> ∫

*𝑓𝑜*(*𝜗, 𝜑*)

properties (stiffness, density) are considered intrinsic, i.e., constants,

Plant-specific chemo-physical fiber properties were collected from

chemical analysis (including acid hydrolysis, chromatography, Kla-

## **Mechanical characterization**

To obtain the models' input properties and to validate them, various mechanical characterization experiments are needed. Macroscopic ones of the biocomposite itself, e.g., three-point bending tests  $(6)$ , are not sufficient, but analyses of the resulting fracture surfaces by SEM already showed the contribution of the fibers to the strength of the biocomposite (MOR>140MPa).